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Specific heat and transport properties of $La_{1-x}Gd_xMnO_3$ at 15 K $\leq T \leq$ 300 K

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1. Introduction

The doped perovskite manganites like $La_{1-x}Gd_xMnO_3 0.0 \le x \le 1.0$ [1] have attracted unprecedented attention due to the occurrence of colossal magnetoresistance (CMR) effects [2] and their technological potential. The Gd doped LaMnO₃ have a very rich phase diagram, depending on the doping concentration, temperature and pressure, being either an anti-ferromagnetic insulator, ferromagnetic metal or charge-ordered insulators. These interesting phenomena pose the necessity of understanding the mechanism behind their origin and the thermodynamic and transport properties of doped manganites.

The crystal structure of LaMnO₃ (space group *pnma*) has an orthorhombic structure with four formula units per unit cell. The occurrence of spin and orbital ordered state in them has been understood in terms of the cooperative Jahn–Teller distortion [3].

In this communication, we have predicted the specific heat and transport properties of $La_{1-x}Gd_xMnO_3$ (0.0 $\leq x \leq 1.0$) as a function of temperature 15 K $\leq T \leq 300$ K using the Modified Rigid Ion Model [4], probably, for the first time. The formalism of the model is described in Section 2 and the results of its application are presented and discussed in Section 3.

ABSTRACT

We have investigated the specific heat and transport properties of the doped perovskite manganites $La_{1-x}Gd_xMnO_3$ with $0.0 \le x \le 1.0$ in the temperature range 15 K $\le T \le 300$ K using the Modified Rigid Ion Model (MRIM), probably, for the first time. At room temperature, for all doping levels the orthorhombic 'O' phase is indicative of a strong Jahn–Teller distortion. Our theoretical results on the specific heat for $La_{1-x}Gd_xMnO_3$ (x = 0.0, 0.75 and 1.0) as a function of temperature 15 K $\le T \le 300$ K are closer to the available experimental data with minor deviations at lower and higher temperature range 30 K $\le T \le 126$ K and revealed new anomalies for x = 0.25 and 0.50 in the temperature range 50 K $\le T \le 100$ K. Further, we have computed the temperature and composition (x) dependence of the transport properties, like the cohesive energy (ϕ), Restrahalen frequency (v_0), Debye temperature (θ_D) and Grüneisen parameter (γ) and compared them with the experimental data available only at room temperature (300 K) for the concentrations x = 0.0 and 1.0.

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2. Essential formalism of modified RIM

We have modified the framework of rigid ion model (RIM) [4] by including the effects of short-range (SR) Hafemeister–Flygare (HF) type [5] overlap repulsion operative upto the second neighbour ions and van der Waals (vdW) attraction due to dipole–dipole (d–d) and dipole–quadrupole (d–q) interactions [6]. The total potential of the Modified RIM (MRIM) is expressed as

$$\phi = -\frac{e^2}{2} \sum_{kk'} Z_k Z_{k'} r_{kk'}^{-1} + n b_1 \beta_{kk'} \exp\{(r_k + r_{k'} - r_{kk'})/\rho_1\} + \frac{n'}{2} b_2 [\beta_{kk} \exp\{(2r_k - r_{kk})/\rho_2\} + \beta_{k'k'} \times \exp\{(2r_{k'} - r_{k'k'})/\rho_2\}] - \sum_{kk'} c_{kk'} r_{kk'}^{-6} - \sum_{kk'} d_{kk'} r_{kk'}^{-8}.$$
 (1)

Here, k(k') denotes the positive (negative) ions and the sum is taken over all (kk') ions. $\beta_{kk'}$ are the Pauling coefficients

$$\beta_{kk'} = 1 + (z_k/n_k) + (z_{k'}/n_{k'}) \tag{2}$$

with $z_k(z_{k'})$ and $n_k(n_{k'})$ as the valence and number of electrons in the outermost orbits of k(k') ions and $r_{kk'}$ and $r_{kk}(=r_{k'k'})$ are the first and second neighbour separations, respectively. In Eq. (1), the first term represents the long-range (LR) Coulomb attraction, the second and third terms are SR HF type repulsion [5] between



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Table 1

Model parameters and	transport	properties of L	$a_{1-x}Gd_xMnO_3$
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Conc. (<i>x</i>)	Temp (K)	Model parameters				Transport properties			
		Mn–O ρ_1 (Å)	Mn-O $b_1(10^{-12} \text{ erg})$	La/Gd–O ρ_2 (Å)	La/Gd-O $b_2(10^{-12} \text{ erg})$	ϕ (eV)	$\upsilon_0~(\text{THz})$	$\Theta_D(\mathbf{K})$	γ
	15	0.4760	0.9926	0.6411	1.5648	-141.81	8.899	426.939	2.231
	65	0.4751	0.9879	0.6400	1.5615	-141.85	8.917	427.763	2.238
	115	0.4742	0.9831	0.6388	1.5582	-141.89	8.935	428.584	2.245
0	165	0.4733	0.9784	0.6377	1.5549	-141.94	8.942	429.403	2.252
	215	0.4724	0.9737	0.6365	1.5516	-141.98	8.967	430.219	2.259
	265	0.4715	0.9691	0.6354	1.5483	-142.02	8.984	431.033	2.266
						-142.05	8.990	431.601	2.271
	300	0.4709	0.9658	0.6346	1.5460	$(-141.81)^{a}$	(8.58) ^b	(388) ^c	(2-3) ^d
	15	0.5102	1.1766	0.6819	1.6952	-143.00	9.021	434.343	2.372
	65	0.5093	1.1714	0.6807	1.6917	-143.05	9.103	435.252	2.374
	115	0.5083	1,1662	0.6795	1.6882	-143.10	9.125	435,158	2.376
0.25	165	0.5074	1.1611	0.6783	1.6847	-143.15	9.141	436,176	2.378
	215	0.5065	1.1560	0.6771	1.6812	-143.21	9.164	438.661	2.380
	265	0.5055	1.1509	0.6759	1.6777	-143.26	9.182	438.858	2.382
	300	0.5049	1.1473	0.6750	1.6753	-143.29	9.194	439.485	2.384
	15	0.5467	1.3800	0.7292	1.8305	-144.91	9.201	440.282	2.440
	65	0.5457	1.3744	0.7279	1.8268	-144.97	9.214	441.290	2.446
	115	0.5447	1.3689	0.7267	1.8231	-144.03	9.223	442.294	2.449
0.50	165	0.5437	1.3633	0.7255	1.8194	-144.09	9.245	443.295	2.456
	215	0.5427	1.3578	0.7242	1.8157	-144.15	9.264	444.293	2.458
	265	0.5418	1.3522	0.7230	1.8120	-144.21	9.281	445.287	2.464
	300	0.5411	1.3484	0.7221	1.8094	-144.25	9.291	447.981	2.468
	15	0.5854	1.6030	0.7792	1.9761	-145.45	9.301	450.192	2.521
	65	0.5844	1.5970	0.7779	1.9722	-145.52	9.310	450.829	2.526
	115	0.5834	1.5911	0.7766	1.9683	-145.59	9.321	450.962	2.530
0.75	165	0.5823	1.5851	0.7753	1.9644	-145.65	9.328	450.981	2.536
	215	0.5813	1.5792	0.7740	1.9605	-145.72	9.331	450.988	2.540
	265	0.5803	1.5733	0.7727	1.9566	-145.79	9.337	450.998	2.544
	300	0.5796	1.5691	0.7718	1.9539	-145.84	9.339	451.315	2.548
	15	0.6226	1 00 20	0.9207	2 1572	146 717	0.400	455 500	2 676
	15	0.0320	1.0000	0.8397	2.1573	- 140./ 1/	9.492	455,509	2.070
	05	0.0318	1.8774	0.8383	2.1532	-146./65	9.509	450.375	2.080
	115	0.6307	1.8710	0.8370	2.1490	- 146.813	9.525	457.160	2.697
1.0	105	0.6296	1.8646	0.8357	2.1449	- 146.861	9.541	457.943	2.707
	215	0.0280	1.6562	0.8343	2.1407	- 146.908	9.558	458.725	2.718
	265	0.6274	1.8519	0.8330	2.1300	-146.956	9.574	459.506	2.728
	200	0.0007	10474	0.0001	2 4 2 2 7	146.000	0.500	460.05 I	2.735
	300	0.6267	1.84/4	0.8321	2.1337	-146.989	9.589	(4/4)	(2-3)
^a Ref. [17	7].								

^b Ref. [18].

^c Ref. [19].

^d Ref. [20].

e Refs. [1,9].

the first and second neighbour ions, respectively. The last two terms are vdW attraction due to d-d and d-q interactions. $c_{kk'}$ and $d_{kk'}$ are the vdW coefficients whose values are determined from the following expressions derived from the Slater-Kirkwood variational method [7]:

$$c_{kk'} = (3e\hbar\alpha_k\alpha_{k'}/2m)[(\alpha_{k/N_k})^{1/2} + (\alpha_{k'/N_{k'}})^{1/2}]^{-1}$$
(3)

$$d_{kk'} = (27e\hbar^2 \alpha_k \alpha_{k'}/8m)[(\alpha_{k/N_k})^{1/2} + (\alpha_{k'/N_{k'}})^{1/2}]^2 \times [(\alpha_{k/N_k})^{1/2} + 20/3(\alpha_k \alpha_{k'/N_k N_{k'}})(\alpha_{k'/N_k})]^{-1}$$
(4)

where *m* and *e* are the mass and charge of the electrons, respectively. $\alpha_k(\alpha_{k'})$ are the polarizabilities of k(k') ions; $N_k(N_{k'})$ are the effective number of electrons responsible for the polarization of k(k') ions.

The model parameters (hardness (b) and range (ρ)) are determined from the equilibrium condition:

$$[d\phi(r)/dr]_{r=r_0} = 0$$
(5)

and the isothermal bulk modulus: . .

$$B_T = (9Kr_0)^{-1} [d^2 \phi(r)/dr^2]_{r=r_0}.$$
(6)

~

Here, K is the crystal structure constant and r_0 is the equilibrium interionic separation.

The calculated values of the model parameters (b_1, ρ_1) and (b_2, ρ_2) are depicted in Table 1 for Mn–O and La/Gd–O bonds, respectively.

The specific heat of $La_{1-x}Gd_xMnO_3$ is calculated at different temperatures (T) and concentrations (x) using the well-known expression [7]:

$$C_V = 9R \left(\frac{T}{\theta_D}\right)^3 \int_0^{\frac{\theta_D}{T}} \frac{e^x x^4}{e^x - 1} dx.$$
(7)

Besides, we have computed the cohesive energy (ϕ) , Restrahalen frequency (v_0), Debye temperature (θ_D) and Grüneisen parameter (γ) of La_{1-x}Gd_xMnO₃ using their expressions reported in our earlier papers [6] and the model parameters listed in Table 1. The computed results are presented and discussed below.

3. Results and discussion

The values of input data (r_0, B_T) for the evaluation of model parameters $(b_1, \rho_1 \text{ and } b_2, \rho_2)$ corresponding to Mn–O and La/Gd–O bonds at different compositions (x) and temperatures (T) are obtained from the well-known Vegard's law using their experimental data [1,8-16] of the end point materials LaMnO₃ and



Fig. 1. The specific heat of LaMnO₃, where the solid line (-) and the squares (\blacksquare) are the present theoretical and experimental [21] results, respectively.

GdMnO₃. Using these input data and vdW coefficients evaluated for La_{1-x}Gd_xMnO₃, we have computed the model parameters $(b_1, \rho_1; b_2, \rho_2)$ using Eqs. (5) and (6) for different compositions (x = 0.0, 0.25, 0.50, 0.75 and 1.0) and temperatures (15 K $\leq T \leq$ 300 K) and listed them in Table 1.

Using the values of these model parameters, we have computed the cohesive energy (ϕ) for La_{1-x}Gd_xMnO₃ (0.0 $\leq x \leq$ 1.0). It is seen from Table 1 that the magnitude of the cohesive energy increases systematically from $\phi = -142.05$ eV for x = 0.0(i.e. LaMnO₃) to -146.989 eV for x = 1.0 (i.e. GdMnO₃). Also, our values of the cohesive energy for pure LaMnO₃ (x = 0.0) is in good agreement with the experimental data [17] available only for x = 0.0 at room temperature. The negative values of cohesive energy show that the stability of the compound is intact. The values of Restrahalen frequency (v_0) (see Table 1) are increasing with Gd concentration (x) and temperature (T), which shows that the bond is becoming stronger with Gd doping. The hardness (b_1, b_2) and the stability $(-\phi)$ of the material are also increasing with the increase of x. Our calculated Restrahlen frequency (8.99 THz) for LaMnO₃ (x = 0.0) are closer to the experimental data (8.58 THz) [18] available only at 300 K.

The calculated values of Debye temperature (θ_D) for LaMnO₃ (at x = 0.0) and GdMnO₃ (at x = 1.0) are almost in good agreement with their corresponding experimental data ($\theta_D = 388$ K) for LaMnO₃ and ($\theta_D = 474$ K) [1,9,19] for GdMnO₃ available only at 300 K. The higher values of Debye temperature indicate the presence of higher phonon frequencies in these manganites. The calculated values of Debye temperature of La_{1-x}Gd_xMnO₃ (for x = 0.0-1.0) at temperature 15 K $\leq T \leq 300$ K are 426.93–460.05 K, which seems within the range of Debye temperature (300–550 K) often found in perovskite manganites. Our calculated Grüneisen parameters (γ) are increasing with composition (x) from 0 to 1.0 and temperature range 15 K $\leq T \leq 300$ K and its values lie in between 2 and 3 as reported by Dia et al. [20].

The specific heat (C) values calculated for $\text{La}_{1-x}\text{Gd}_x\text{MnO}_3$ in the temperature range 15 K $\leq T \leq 300$ K for Gd concentrations $0.0 \leq x \leq 1.0$ are displayed in Figs. 1–3 and compared with the available experimental data [1,21]. It is noticed from Fig. 1 that temperature variations of calculated specific heat for undoped LaMnO₃ is almost in fair agreement with the experimental data [21] in the temperature range 50 K $\leq T \leq 125$ K. Such a close match between theoretical and experimental [21] results are remarkable in a LaMnO₃ system which shows the orbital, lattice, magnetic and strong correlation effects. The deviations at higher temperature range 150 K $\leq T \leq 300$ K might be due to the exclusion



Fig. 2. ((a), (b), (c)). The specific heat of $La_{0.75}Gd_{0.25}MnO_3$, $La_{0.50}Gd_{0.50}MnO_3$ and $La_{0.25}Gd_{0.75}MnO_3$, where the solid lines (–) and the squares (\blacksquare) are the present theoretical and experimental [1] results, respectively.



Fig. 3. The specific heat of GdMnO₃, where the solid line (-) and the squares (\blacksquare) are the present theoretical and experimental [1] results, respectively.

of anharmonic effects in the MRIM framework. The Fig. 1 also depicts that the undoped LaMnO₃ undergoes the transition from paramagnetic to ferromagnetic state at 126 K as is indicated by a peak in the specific heat curve. This feature is revealed from both the present theoretical and experimental [21] results. On Gd doping (for x = 0.25, 0.50 and 0.75) in LaMnO₃, the specific heat increases systematically with temperature as is seen from Fig. 2((a), (b), (c)). Our theoretical results on the specific heat for La_{0.75}Gd_{0.25}MnO₃, La_{0.50}Gd_{0.50}MnO₃ and La_{0.25}Gd_{0.75}MnO₃ have shown an anomaly around 60–70 K (see Fig. 2((a), (b), (c)) which might be the similar magnetic transitions as exhibited by the undoped LaMnO₃ (x = 0.0) and GdMnO₃ (x = 1.0) compounds (see Figs. 1 and 3).

It is seen from Figs. 2(c) and 3 that a cusp-like anomaly occurs at 40–60 K. These features correspond to the transition from paramagnetic to canted antiferromagnetic (CA) state. A shoulder is displayed in Figs. 2(c) and 3 by the specific heat (C) curve below T_{CA} and the sharp increase towards the low temperatures exhibited by both the theoretical and experimental [1] results reveal the polarization of Gd spins within the effective field of Mn sublattice, but no evidence for a further magnetic transition of Mn sublattice into an incommensurate phase was found. It is interesting to note from Figs. 2(c) and 3 that our specific heat results are almost in good agreement with the experimental data [1] in the range 25 K $\leq T \leq$ 75 K, but they show remarkable deviations at lower temperatures 0 K $\leq T \leq 25$ K in both the manganites. These deviations might be partly due to the exclusion of the effects of zero-point energy and the change in entropy in the framework of the MRIM. Our results on the specific heat at x = 0.25 and 0.50 could not be compared due to the lack of experimental data, hence they will serve as a guide to the experimental workers in future.

4. Conclusion

On the basis of the overall discussions it may be concluded that the specific heats (Figs. 1–3) at lower and higher temperatures show a significant change and many interesting features. The slight deviations in the specific heat values (Figs. 1–3) between the low and high temperatures are, probably, due to the dominant role played by the Jahn–Teller (JT) effect in La_{1–x}Gd_xMnO₃ in this temperature range. Here, it is noteworthy that a quantitative estimate of the JT effect would make the results of study more useful. It is also interesting to note that our results on temperature dependence of specific heat have revealed an anomaly at lower temperatures for all the compositions (*x*). These anomalies have shifted to the lower temperatures with increasing Gd doping for $0.0 \le x \le 1.0$. This might be due to the strong Jahn–Teller distortion at lower temperatures.

The description of the specific heat and transport properties for $La_{1-x}Gd_xMnO_3$ system seems remarkable in view of the inherent simplicity of the Modified Rigid Ion Model. Our theoretical results on these properties at lower and higher temperatures can be improved further by incorporating the effects of ferromagnetic spin wave contributions, zero-point energy, change in entropy and anharmonicity in the framework of the MRIM

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